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Navy Case No. 77,897

Commissioner of Patents and Trademarks  
Washington, D. C. 20231

Applicant: Imam et al.

For: "POROUS METAL/ORGANIC POLYMERIC COMPOSITES"

Sir:

Transmitted herewith are the papers above-identified constituting a Patent Application filed by the Department of the Navy on behalf of the above-named Applicant(s).

The total filing fee has been computed in accordance with the following formula:

Basic Fee ----- \$ 770.00

Additional Fees:

1. Total number of claims in excess  
of 20 times \$22.00-----\$44.00

2. Number of independent claims  
minus 3 times \$78.00-----\$ -0-

Total Filing Fee----- \$ 814.00

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Respectfully,

THOMAS E. McDONNELL  
Reg. #26950

Enclosures:

(1) Application Papers

Navy Case No. 77,897

## APPLICATION FOR LETTERS PATENT

TO ALL WHOM IT MAY CONCERN:

BE IT KNOWN THAT M. Ashraf Imam, Bhakta B. Rath and Teddy M. Keller are citizens of the United States of America, and residents of Great Falls, Va., Oakton, Va., and Alexandria, VA., have invented certain new and useful improvements in "POROUS METAL/ORGANIC POLYMERIC COMPOSITES" of which the following is a specification:

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A  
**ORIGINAL**

1  
2  
3                   **POROUS METAL/ORGANIC POLYMERIC COMPOSITES**

4  
5                   **Background of the Invention**

6                   **1. Field of the Invention**

7                   The present invention relates generally to metal composites  
8                   and more particularly to metal/polymer composites.

9  
10                  **2. Description of the Background Art**

11                  In many cases, such as in factories, motor vehicles, ships,  
12                  and submarines, it is desirable to absorb as much noise as  
13                  possible. In factories, absorbance of machinery noises can  
14                  minimize noise pollution caused by industrial sounds. In motor  
15                  vehicles, absorbance of engine noises and noises from road  
16                  vibration enhances driver and rider comfort. In submarines and  
17                  ships, the absorbance of machinery sounds helps prevent detection  
18                  and identification by unfriendly forces. Additionally, in regions  
19                  where earthquakes are common, the structural components of  
20                  buildings, bridges, and similar artifacts must not only be strong,  
21                  but should be able to dampen vibrations.

22                  Past noise reduction efforts have previously taken one of two  
23                  approaches. The use of soft coupling components (e.g., bushings,  
24                  pads) to isolate machinery, and the incorporation of structural  
25                  acoustic damping materials.

1           Both of these prior art approaches have had drawbacks. Soft  
2       coupling components add weight to a device, may require  
3       maintenance, and may not be consistent with the required design and  
4       performance features. Currently available acoustic damping  
5       materials do not exhibit adequate acoustic behavior over broad  
6       ranges of temperature, lack the required strain amplitude and  
7       frequency, or do not have the desired strength and environmental  
8       resistance.

9           Significant advances have been made to increase the strength  
10      and stiffness of lightweight materials. Examples of these  
11      developments include the superior density-normalized strength and  
12      stiffness of Al-Li, hard dispersion strengthened aluminum and  
13      titanium, aluminides, and whisker or fiber reinforced aluminum and  
14      magnesium. Although those materials have excellent structural  
15      characteristics, the demand continues for high strength,  
16      lightweight structural materials that also vastly improve damping  
17      capability.

18           United States Patent No. 4,759,000, the entirety of which is  
19      incorporated herein by reference for all purposes, to Ronald P.  
20      Reitz describes acoustically transparent windows made of an  
21      aluminum/nickel foam impregnated with an acoustically non-absorbent  
22      rubber such as BE silicone rubber RTV-11. Nothing in that patent  
23      teaches or suggests that a useful material may be obtained by  
24      impregnating a metal foam with an acoustically absorptive polymer.

## Summary of the Invention

Accordingly, it is an object of this invention to produce a light structural materials having good structural and acoustic damping properties.

It is another object of the present invention to reduce noise derived from the operation of machinery.

It is a further object of the present invention to acoustically isolate internal areas of a structure from their surrounding environment.

These and other objects are achieved by impregnating a metal foam with a polymer.

#### Brief Description of the Drawings

A more complete appreciation of the invention will be readily obtained by reference to the following Description of the Preferred Embodiments and the accompanying drawings in which like numerals in different figures represent the same structures or elements, wherein:

Fig. 1 shows the acoustic damping behavior of an aluminum alloy foam/phthalonitrile composite, heat treated under the various condition described in the accompanying Example 3.

Fig. 2, Fig. 3 and Fig. 4 show the results of damping

1 measurements for a copper foam/phthalonitrile composite, over a  
2 frequency range of 0.1 to 10 Hz.  
3

4 Fig. 5, Fig. 6, and Fig. 7 show the damping behavior of a  
5 titanium foam/phthalonitrile composite under different heat  
6 treatment conditions as described in Example 5.  
7

8 Fig. 8 and Fig. 9 show the damping behavior of a zinc  
9 foam/phthalonitrile composite under different heat treatment  
10 conditions as described in Example 5.  
11

12 Fig. 10 is a graph showing the  $\text{Tan}\delta$  over a frequency range of  
13 0.1 to 10 Hz for various polymers and metal foams.  
14

15 Fig. 11 shows the damping behavior of a white rubber/aluminum  
16 alloy foam over the frequency range of 0.1 to 10 Hz.  
17

18 Fig. 12 shows the damping behavior of a red rubber/aluminum  
19 alloy foam over the frequency range of 0.1 to 10 Hz.  
20

21 Fig. 13 shows the damping behavior of an epoxy/aluminum alloy  
22 foam over the frequency range of 0.1 to 10 Hz.  
23

24 Fig. 14 shows the damping behavior of an acrylic  
25 polymer/aluminum alloy foam over the frequency range of 0.1 to 10

1 Hz.

2  
3 Fig. 15 shows the stress/strain diagram for an aluminum alloy  
4 foam/phthalonitrile composite that was heat treated at 280°C for 18  
5 hours, at 325°C for 4 hours, and at 375°C for 4 hours.

6  
7 Fig. 16 shows the stress/strain diagram for an aluminum alloy  
8 foam/acrylic composite.

9  
10 **Description of the Preferred Embodiments**

11  
12 The metal foam of the present invention may be any porous  
13 metal workpiece, particularly a metal foam, regardless of shape or  
14 percent density, having an open cell structure that permits  
15 impregnation with an uncured polymer. Particularly useful metals  
16 include aluminum, titanium, nickel, copper, iron, zinc, lead,  
17 silver, gold, platinum, tantalum, and alloys (including steel)  
18 based on these metals. Other metals may also be used. Aluminum  
19 and titanium and alloys thereof are particularly useful because of  
20 their low density.

21 Metal foams may be produced by a variety of known methods.  
22 For example, a molten metal may be placed under high pressure so  
23 that it dissolves a non-reactive gas. When the pressure is  
24 released and the metal is allowed to cool, the dissolved gas  
25 escapes and leaves open-celled pores in the metal body. In other

processes, foaming agents are added to molten metal. Another process mixes monomers with metal particles and heats the mixture to induce foaming and solidification of the resin. As used herein, the term "resin" encompasses prepolymers, monomers, and mixtures thereof. A "cured resin" is the cured polymer. With further heating, the metal particles consolidate and the polymer hydrolyzes. United States Patent No. 4,569,821, the entirety of which is incorporated herein by reference for all purposes, improves upon that process by substituting a stabilized hydrogel for the monomers, allowing more complete pyrolysis of the organic components during formation of the metal foam.

In a foam, pore size is defined as the number of pores per linear unit length. Because foams have only thin ligaments between the pores, the void diameter of a pore is approximately the reciprocal of the pore size of the foam. Decreasing the void diameter increases the contact surface area between the metal foam and the polymer. Also, as the percent foam density (*%Foam Density = Density of a One Unit Volume of Foam/Density of One Unit Volume of Metal x 100*) of the metallic foam decreases, the final product increasingly exhibits the structural characteristics of the polymer material. Thus, as the percent foam density of the metallic foam decreases, the stiffness of the composite decreases. Even with extremely low percent foam densities, however, the stiffness of the composite is greater superior to that of either of the individual components. Percent foam density and pore size also determine the

1 number of polymer/metal interfaces that an acoustic vibration must  
2 traverse. As explained below, the polymer/metal interfaces  
3 contribute mainly to the sound damping capabilities of the  
4 composites of the present invention. A typical useful pore size is  
5 5-100 pores/inch. More often, a pore size of about 10-50 pores/inch  
6 is used, and most often a pore size of about 10-40 pores/inch is  
7 used. Typically, the percent foam density is about 5-40. More  
8 often, the percent foam density is about 8-10.

9       Typically, to improve the predictability of the structural and  
10 acoustic properties of the composite products of the present  
11 invention, the pores (also referred to in the present specification  
12 and claims as "cells") within the metal foam have a locally uniform  
13 size and distribution. Throughout the present specification and  
14 claims, a foam has a locally uniform size and distribution of pores  
15 if most of the pores of the foam are surrounded by evenly  
16 distributed pores having approximately the same void diameter as  
17 the surrounded pore. Local non-uniformity in the size or  
18 distribution of the pores within the metal foam decreases the  
19 ability to predict the characteristics of the resulting composite.  
20 If desired, the metal foam may be divided into regions of different  
21 pore size, or may have a gradation of pores sizes in any direction  
22 along the metal foam, while maintaining locally uniform pore size,  
23 without harming the predictability of performance.

24       The metal foam may be impregnated by any available method.  
25 Typically, the metal foam is impregnated by contacting it with a

resin component. The resin component may be a neat resin or a neat blend of resins, or may include any catalysts, curing agents, or additives desired. The resin component may be a powder (of sufficiently small particle size to penetrate the pores of the metal foam), a melt, a room temperature liquid, or a solution, and may include mixtures of several prepolymers and/or monomers. A vacuum or positive pressure may be applied to assist the penetration of the resin component into the metal foam. Solvent, if present, is removed by evaporation. The resin component is then converted (solidified or consolidated by any method, typically heating to polymerize and/or cure the resin, or cooling to solidify a molten resin) to a solid bulk polymer (partially crosslinked, fully crosslinked, or non-crosslinked) that fills or partially fills the open-cells of the metal foam.

The viscosity of the resin component (i.e., the impregnant) can be adjusted both thermally or by adjusting the amount of any solvent used, if necessary. Preferably, the impregnant viscosity should be selected to allow the metal foam to be completely impregnated with the resin component under practical processing conditions. A high impregnant viscosity may restrict the ability of the resin component to completely penetrate the open porous structure of the metal foam. This problem may be overcome by forcing, under positive pressure, the resin component into the pores of the foam. For any given impregnant used, an appropriate impregnant viscosity may be selected empirically, without undue

1 experimentation, given the guidance provided by this specification  
2 and the accompanying examples. Resin component in powder form can  
3 be forced into the pores of the metal foam by any method. For  
4 example, the powdered resin component may be poured on top of the  
5 metal foam, and positive or negative pressure may be applied to the  
6 powder, forcing it into the pores of the foam. At the same time,  
7 the metal foam may be vibrated to aid in impregnation.

8 The polymeric component of the present invention is typically  
9 selected to have high intrinsic acoustic damping. Basically, the  
10 acoustic damping ability of a polymer is determined by its dynamic  
11 modulus at a given frequency. In the case of rubber materials, the  
12 acoustic damping ability of unhardened rubbers is significantly  
13 greater than that of hardened or fully hardened rubbers. In  
14 general, thermoplastics and thermosets, unlike elastomers, provide  
15 excellent results when used according to the presently claimed  
16 invention, whether cured or uncured.

17 Polymers particularly useful in the present invention include  
18 phthalonitriles, epoxies, acrylics, silicones, polyurethanes,  
19 polyimides, polyvinyls, polycarbonates, natural rubbers, synthetic  
20 rubbers, phenolics, polyolefins, polyamides, polyesters,  
21 fluoropolymers, poly(phenylene ether ketones), poly(phenylene ether  
22 sulfones), poly(phenylene sulfides) and melamine-formaldehyde  
23 resins.

24 The acoustic damping capabilities of the composite of the  
25 present invention arise in part from the acoustic properties of the

1 polymerized resin component and in part from dissipation of energy  
2 at the polymer/metal interface. Energy is never transferred  
3 without loss at interfaces between different materials. Therefore,  
4 as the number of interfaces that an acoustic vibration must  
5 traverse increases, the percentage of dissipated acoustic energy  
6 also increases. A metal foam/polymer composite provides numerous  
7 interfaces between the polymer and the metal matrix.

8 Additionally, relative motion between the polymer and metal at  
9 their interfaces increases the loss of acoustic energy at these  
10 interfaces. Thus, the acoustic damping properties of a  
11 polymer/metal matrix composite may be improved, in some cases, by  
12 forming the composite under conditions that avoid or minimize  
13 chemical or bonding between the metal and the polymer component.  
14 Bonding between the polymeric component and the interface may be  
15 controlled by the combination of metal and polymeric agent  
16 selected, the choice of curing agent and/or curing mechanism, and  
17 the application of a release agent, if any, to the metal foam  
18 before impregnation with resin component. In other cases, however,  
19 vibration of a bond between a metal foam and a polymer may be a  
20 loss mechanism that increases acoustical damping across the desired  
21 frequency range.

22 For acoustic damping applications, the composite of the  
23 present invention should provide a sufficient number of  
24 polymer/metal interfaces to allow good acoustic damping. Thus,  
25 particularly when formed as a sheet for acoustic damping

1 applications, the smallest dimension of the metal foam (for a  
2 sheet, the thickness) is usually at least about 1.5 times the  
3 average void diameter of the metal foam. In many cases, for  
4 acoustic damping applications, the smallest dimension of the metal  
5 foam will be three or more times greater than the average void  
6 diameter of the metal foam.

7 Also, acoustic damping in these materials might be further  
8 improved by the inclusion of polymer/gas and/or gas/metal  
9 interfaces, which further increase the number of interfaces that an  
10 acoustic vibration must traverse through the composite. These  
11 polymer/gas and/or gas/metal interfaces may be formed by many  
12 mechanisms, such as the use of a foamed resin component to produce  
13 the polymeric component of the composite, the dissolution of the  
14 neat resin or blend of resins in a solvent before impregnation, or  
15 the inclusion of minute amounts of gas, or materials that form a  
16 gas or a vapor during curing, within the resin component used to  
17 form the polymeric component. The amount of gas or gas forming  
18 materials should be sufficient to significantly enhance the  
19 acoustic damping capabilities of the composite, but should not  
20 provide sufficient gas to essentially destroy the structural  
21 integrity of the composite and/or the polymeric component thereof.  
22 The optimum amount of any gas or gas producing agent used in the  
23 resin component will vary depending upon the desired use for the  
24 composite and may be empirically determined without undue  
25 experimentation.

1           The interaction of the metal foam and the polymer matrix also  
2           contributes to the structural strength of the composite.  
3           Consequently, the structural strength of the metal foam/polymer  
4           composite is greater than the individual structural strengths of  
5           the metal foam and the polymer.

6           Composites according to the present invention may be  
7           constructed to exhibit acoustic damping across a frequency band  
8           residing within a range of typically about 0.001-80 kHz. The  
9  
10          precise frequency band, as well as the bandwidth over which a  
11          composite according to the present invention exhibits acoustic  
12          damping, is determined by the selected polymer, metal, pore size  
13          and percent foam density. The frequency range of damping may be  
14          extending by stacking together metal foam/polymer composites having  
15          different pore sizes, percent metal foam densities, polymers and/or  
16          metals. Several sheets may be laminated together with adhesive, or  
17          in some cases will bond to each other during curing of one or more  
18          of the polymers within the metal foams, to form a laminate. Pore  
19          size and percent metal foam density may also be varied, without  
20          stacking, by providing a metal foam having regions of different  
21          average pore size or having an average pore size that is graded in  
22          one or more directions. It may also be possible to vary the  
23          polymer used within a single sheet. For example, a foam may be  
24          impregnated with a first resin component. The resin component-  
25          impregnated foam may then be subjected to pressure on, for example  
               its upper surface, to force any portion of the first resin

1 component out in the upper portion of the metal foam down into the  
2 lower portion. The resin component-impregnated foam may then be  
3 solidified while under this positive pressure. After partial or  
4 complete consolidation or solidification of the first resin  
5 component in the lower portion of the metal foam, the upper portion  
6 of the metal foam may be impregnated with a second resin component.

7 Aside from enhanced performance, components manufactured using  
8 composites according to the present invention may alter the noise  
9 properties of machinery, aerospace vehicles, domestic vehicles,  
10 military vehicles, commercial vehicles, marine vehicles and  
11 maritime vehicles. Also, composites according to the present  
12 invention are readily manufactured, for example, by resin transfer  
13 molding (RTM), resin infusion molding, or resin injection molding.

14  
15 Having described the invention, the following examples are  
16 given to illustrate specific applications of the invention  
17 including the best mode now known to perform the invention. These  
18 specific examples are not intended to limit the scope of the  
19 invention described in this application.

20  
21 EXAMPLES  
22

23 Example 1: Preparation of phthalonitrile prepolymer

24 Phthalonitrile monomer, 4,4'-bis(3,4-dicyanophenoxy)biphenyl,  
25 was purchased from Daychem Laboratories. 10 g of the monomer was

placed in an aluminum planchet and melted on a hot plate at 250°C (monomer melts around 235°C). The monomer melt was degassed for about 2h to eliminate trace amounts of solvent present. The phthalonitrile prepolymer was synthesized by adding 0.15- 0.168 g (1.5-1.68 wt%) of 1,3-bis(3-aminophenoxy)benzene, obtained from National Starch Corporation, to the monomer melt. The melt was stirred for 15 min. and was used for fabrication of phthalonitrile/metal foam composite specimens.

Example 2: Fabrication of phthalonitrile/aluminum foam composite

An aluminum mold, 2"x0.6"x0.2", coated with a teflon mold release agent was used for fabrication of composite specimens. Two aluminum foam strips (1"x0.5"x0.185", density 6-8 % of the solid material, pore size - 40 pores per inch (ppi)) separated by a teflon film, were placed in the mold and heated to 250°C. Approximately 2-3 g of the prepolymer melt synthesized as described above in Example 1 with 1.68 wt% curing additive was poured over the metal foam and degassed for about 15 min. with periodic venting to ensure a good flow of the resin throughout the metal foam. The mold was then heated in an air circulating oven for 9h at 280°C and cooled back to room temperature over a 3h span. The composite samples made with this prepolymer showed an incomplete penetration of the resin into the metal foam. Therefore, subsequent composite fabrications involved a slower curing prepolymer made with 1.5% curing additive. The prepolymer has an initial lower viscosity.

1 Example 3: Fabrication of phthalonitrile/aluminum foam composite  
2 An aluminum mold, 2"x0.6"x0.2", coated with a teflon mold  
3 release agent was used for fabrication of composite specimens. Two  
4 aluminum foam strips (1"x0.5"x0.185", density 6-8 % of the solid  
5 material, pore size - 40 ppi) separated by a teflon film, were  
6 placed in the mold and heated to 250°C. Approximately 2-3 g of the  
7 prepolymer melt synthesized with 1.5 wt% curing additive was poured  
8 over the metal foam and degassed for about 15 min. with periodic  
9 venting to ensure a good flow of the resin throughout the metal  
10 foam. The mold was then heated in an air circulating oven for 9h  
11 at 280°C and cooled back to room temperature over a 3h span. The  
12 composite samples made with this prepolymer showed a complete  
13 penetration of the resin through the metal foam. The mechanical  
14 and damping properties of the phthalonitrile/aluminum composite  
15 samples were evaluated after heat treatment at the following  
16 conditions: (A) 9h at 280°C (B) 18h at 280°C (C) 18h at 280, 4h at  
17 325°C and (D) 18h at 280, 4h at 325, 4h at 375°C. Conditions (C)  
18 and (D) employed an inert atmosphere purge of argon.

19  
20 Example 4: Fabrication of phthalonitrile/copper foam composite  
21 An aluminum mold, 2"x0.6"x0.2", coated with a teflon mold  
22 release agent was used for fabrication of composite specimens. Two  
23 copper foam strips (1"x0.5"x0.185", density 6-8 % of the solid  
24 material, pore size - 10 ppi), separated by a teflon film, were  
25 placed in the mold and heated to 250°C. The prepolymer melt

1 synthesized as described above in Example 1 with 1.5 wt% curing  
2 additive was poured over the metal foam and degassed with periodic  
3 venting to ensure a good flow of the resin throughout the metal  
4 foam. The mold was then heated in an air circulating oven for 9h  
5 at 280°C and cooled back to room temperature over a 3h span. The  
6 mechanical and damping properties of the phthalonitrile/copper  
7 composite samples were evaluated after heat treatment at the  
8 following conditions: (A) 9h at 280°C (B) 18h at 280°C (C) 18h at  
9 280, 4h at 325°C and (D) 18h at 280, 4h at 325,4h at 375°C.  
10 Conditions (C) and (D) employed an inert atmosphere purge of argon.

11  
12 Example 5: Fabrication of phthalonitrile/titanium foam composite

13 An aluminum mold (2"x0.6"x0.2", density 6-8 % of the solid  
14 material, pore size - 10 ppi) coated with a teflon mold release  
15 agent was used for fabrication of composite specimens. Two  
16 titanium foam strips, 1"x0.5"x0.185"x0.185", separated by a teflon  
17 film, were placed in the mold and heated to 250°C. The prepolymer  
18 melt synthesized as described above in Example 1 was poured over  
19 the metal foam and degassed for about 15 min. with periodic venting  
20 to ensure a good flow of the resin throughout the metal foam. The  
21 mold was then heated in an air circulating oven for 9h at 280° C  
22 and cooled back to room temperature over a 3h span. The mechanical  
23 and damping properties of the phthalonitrile/titanium composite  
24 samples were evaluated after heat treatment at the following  
25 conditions: (A) 9h at 280°C (B) 18h at 280°C (C) 18h at 280, 4h at

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1       325°C and (D) 18h at 280, 4h at 325, 4h at 375°C. Conditions (C)  
2       and (D) employed an inert atmosphere purge of argon.  
3  
4

4       Example 6: Fabrication of phthalonitrile/zinc foam composite

5       An aluminum (mold, 2"x0.6"x0.2", density 6-8 % of the solid  
6       material, pore size - 10 ppi) coated with a teflon mold release  
7       agent was used for fabrication of composite specimens. Two zinc  
8       foam strips, 1"x0.5"x0.185", separated by a teflon film, were  
9       placed in the mold and heated to 250°C. The prepolymer melt  
10      synthesized as described above in Example 1 with 1.5% curing  
11      additive was poured over the metal foam and degassed for about 15  
12      min. with periodic venting to ensure a good flow of the resin  
13      throughout the metal foam. The mold was then heated in an air  
14      circulating oven for 9h at 280°C and cooled back to room  
15      temperature over a 3h span. The mechanical and damping properties  
16      of the phthalonitrile/zinc composite samples were evaluated after  
17      heat treatment at the following conditions : (A) 9h at 280°C (B)  
18      18h at 280°C (C) 18h at 280, 4h at 325°C and (D) 18h at 280, 4h at  
19      325, 4h at 375°C. Conditions (C) and (D) employed an inert  
20      atmosphere purge of argon.  
21  
22

22      Example 7. Fabrication of Aluminum Alloy Foam / Rubber (White)  
23      Composite.

24       An aluminum mold, 2"x0.6"x0.2", coated with a teflon mold  
25      release agent was used for fabrication of composite specimens. Two

1 aluminum foam strips, 1"x0.5"x0.185", separated by a teflon film,  
2 were placed in the mold. The aluminum foam strips were submerged in  
3 the polymeric uncured natural rubber dissolved in an organic  
4 solvent and with a small amount of peroxide to thermally cure the  
5 rubber. The composite was subjected to pressure to remove the  
6 solvent and to consolidate the composite fabrication. The  
7 impregnated uncured rubber/aluminum foam composition was heated at  
8 about 160°C for 1-2 hours to crosslink the rubber.  
9

10 Example 8. Fabrication of Aluminum Alloy Foam /Red Rubber  
11 Composite.

12 An aluminum mold, 2"x0.6"x0.2", coated with a teflon mold  
13 release agent was used for fabrication of composite specimens. Low  
14 viscosity uncured RTV silicone rubber was quickly placed in the  
15 mold and two aluminum foam strips, 1"x0.5"x0.185", separated by a  
16 teflon film, were submerged in the uncured rubber. The mold was  
17 then placed in a vacuum oven at reduced pressure for 15 minutes.  
18 The rubber was then cured at room temperature for 24 hours  
19 affording a rubber/aluminum foam composite.  
20

21 Example 9. Fabrication of Epoxy/Aluminum Alloy Foam Composite.

22 An aluminum mold, 2"x0.6"x0.2", coated with a teflon mold  
23 release agent was used for fabrication of composite specimens. Epon  
24 828 and an aromatic diamine were mixed thoroughly and placed into  
25 the aluminum mold. Several aluminum foam strips, 1"x0.5"x0.185",

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1 separated by a teflon film, were submerged in the epoxy/amine  
2 composition at approximately 100°C in a vacuum oven. At this time,  
3 a vacuum was applied to consolidate the dispersion of the  
4 composition into the pores of the aluminum foam. The composition  
5 was cured by heating at 65°C for 5-6 hours. The epoxy/aluminum foam  
6 composite was used for evaluating damping and mechanical  
7 properties.

8

9 Example 10. Fabrication of Acrylic/Aluminum Alloy Foam Composite.

10 An aluminum mold, 3"x 2" was used for fabrication of composite  
11 specimens. 18 ml of catalyst B was added to 40 ml of resin A of  
12 the EPO-KWICK components and thoroughly mixed. The mixture was  
13 poured on top of several aluminum foam strips, 1"x0.5"x0.185",  
14 contained in the aluminum mold. At this time, a vacuum was applied  
15 for 10 minutes to degas and consolidate the dispersion of the  
16 composition into the pores of the aluminum foam. The composition  
17 was cured by keeping at room temperature overnight. The  
18 acrylic/aluminum alloy foam composite was used for evaluating  
19 damping and mechanical properties.

20

21 Example 11. Preparation of Porous Metal/Organic Polymeric Composite  
22 Samples for Damping Measurements.

23 For the purpose of demonstration, we have used selected foams  
24 and filler materials. Many metals or even high strength alloys can  
25 be used as skeleton material and also wide variety of filler

1 materials can be used to design the composite possessing required  
2 strength and damping capability. Other factors such as  
3 environmental compatibility, temperature and chemical compatibility  
4 in addition to cost can dictate the choice of the materials to be  
5 used.

6  
7 The composites produced in Examples 3-9 were used for damping  
8 measurements. The samples were prepared with dimensions having 4.5  
9 mm thickness, 10 mm width, and 32 mm length. The damping capacity  
10 of the samples were measured with a dynamic mechanical thermal  
11 analyzer (DMTA). A small sinusoidal mechanical stress is applied to  
12 the sample and the resulting sinusoidal strain transduced.  
13 Comparison of the amplitude of the signals yields the complex  
14 dynamic modulus  $E^*$ . The phase lag ( $\delta$ ) of strain behind stress is  
15 measured and the storage modulus and loss factor of the material  
16 are calculated.

17  
18 Example 12. Damping Characteristics of Aluminum Alloy  
19 Foam/Phthalonitrile Composite.

20 The aluminum alloy foam/phthalonitrile composite prepared in  
21 Example 3 was evaluated for its damping characteristics under 1 Hz  
22 frequency. The results of the damping measurements were shown in  
23 Fig. 1. The figure shows the damping behavior of the composite  
24 under different heat treatment conditions as described in Example  
25 3. The damping measurements are plotted with temperature. It is

1 noted that damping peak locations at a given temperature can be  
2 adjusted based on heat treatment of composite. The room temperature  
3 damping (flat portion of curve) is also higher than the best  
4 damping material such as Vacrosil™ as shown in Fig. 10.  
5

6 Example 13. Damping Characteristics of Copper Foam/Phthalonitrile  
7 Composite.  
8

9 The copper foam/phthalonitrile composite prepared in Example  
10 4 was evaluated for its damping characteristics under 0.1 to 10 Hz  
11 frequency. The results of the damping measurements were shown in  
12 Fig. 2, Fig. 3, and Fig. 4. The figures show the damping behavior  
13 of the composite under different heat treatment conditions as  
14 described in Example 4. The damping measurements are plotted with  
15 temperature. It is noted that damping peak locations at a given  
16 temperature can be adjusted based on heat treatment of the  
17 composite. The room temperature damping (flat portion of curve) is  
18 also higher than the best damping material such as Vacrosil™ as  
19 shown in Fig. 10.

20 Example 14. Damping Characteristics of Titanium Foam/Phthalonitrile  
21 Composite.  
22

23 The titanium foam/phthalonitrile composite prepared in Example  
24 5 was evaluated for its damping characteristics under 0.1 to 10 Hz  
25 frequency. The results of the damping measurements were shown in  
Fig. 5, Fig. 6, and Fig 7. The figures show the damping behavior of

1 the composite under different heat treatment conditions as  
2 described in Example 5. The damping measurements are plotted with  
3 temperature. It is noted that damping peak locations at a given  
4 temperature can be adjusted based on heat treatment of the  
5 composite. The room temperature damping (flat portion of curve) is  
6 also higher than the best damping material such as Vacrosil™ as  
7 shown in Fig. 10.

8

9 Example 15. Damping Characteristics of Zinc Foam/Phthalonitrile  
10 Composite.

11 The zinc foam/phthalonitrile composite prepared in Example 6  
12 was evaluated for its damping characteristics under 0.1 to 10 Hz  
13 frequency. The results of the damping measurements were shown in  
14 Fig. 8 and Fig. 9. The figures show the damping behavior of the  
15 composite under different heat treatment conditions as described in  
16 Example 5. The damping measurements are plotted with temperature.  
17 It is noted that damping peak locations at a given temperature can  
18 be adjusted based on heat treatment of the composite. The room  
19 temperature damping (flat portion of curve) is also higher than the  
20 best damping material such as Vacrosil™ as shown in Fig. 10.

21

22 Example 16. Damping Characteristics of Aluminum Alloy Foam/white  
23 rubber Composite.

24 The aluminum alloy foam/white composite prepared in Example 7  
25 was evaluated for its damping characteristics under 0.1 to 10 Hz

frequency. The results of the damping measurements were shown in Fig. 11. This figure shows the damping behavior of the composite. The damping measurements are plotted with temperature. It is noted that damping characteristics are better than the best damping material such as Vacrosil™ as shown in Fig. 10.

Example 17. Damping Characteristics of Aluminum Alloy Foam/red rubber Composite.

The Aluminum Alloy Foam/red rubber composite prepared in Example 8 was evaluated for its damping characteristics under 0.1 to 10 Hz frequency. The results of the damping measurements were shown in Fig. 12. This figure shows the damping behavior of the composite. The damping measurements are plotted with temperature. It is noted that damping characteristics are better than the best damping material such as Vacrosil™ as shown in Fig. 10.

Example 18. Damping Characteristics of Aluminum Alloy Foam/Epoxy Composite.

The Aluminum Alloy Foam/epoxy composite prepared in Example 8 was evaluated for its damping characteristics under 0.1 to 10 Hz frequency. The results of the damping measurements were shown in Fig. 13. This figure shows the damping behavior of the composite. The damping measurements are plotted with temperature. It is noted that damping characteristics are better than the best damping material such as Vacrosil™ as shown in Fig. 10.

1 Example 19. Damping Characteristics of Aluminum Alloy Foam/Acrylic  
2 Composite.

3 The Aluminum Alloy Foam/Acrylic composite prepared in Example  
4 10 was evaluated for its damping characteristics under 0.1 to 10 Hz  
5 frequency. The results of the damping measurements were shown in  
6 Fig. 14. The figure shows the damping behavior of the composite.  
7 The damping measurements are plotted with temperature. It is noted  
8 that damping characteristics are better than the best damping  
9 material such as Vacrosil™ as shown in Fig. 10.

10  
11 Example 20. Mechanical Properties of Aluminum Alloy  
12 Foam/Phthalonitrile Composite.

13 The aluminum alloy foam/phthalonitrile composite prepared in  
14 Example 3 was evaluated for its room temperature compressive  
15 mechanical properties after various heat treatment exposures. For  
16 example, Fig. 15 shows the stress/strain diagram for the composite  
17 which was heat treated at 280°C for 18 hours, at 325°C for 4 hours,  
18 and at 375°C for 4 hours. The figure shows that the composite  
19 exhibit superior mechanical properties relative to the cured  
20 phthalonitrile resin.

21  
22 Example 21. Mechanical Properties of Aluminum Foam/Acrylic  
23 Composite.

24 The aluminum alloy foam/Acrylic composite prepared in Example  
25 10 was evaluated for its room temperature compressive mechanical

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1 properties after various heat treatment exposures. For example,  
2 Fig. 16 shows the stress/strain diagram for the composite. The  
3 figure shows that the composite exhibit superior mechanical  
4 properties relative to the aluminum alloy foam.

5  
6 Additional information concerning the present invention may be  
7 found in the copending United States Patent Application of Imam,  
8 Sastri and Keller, entitled LIGHTWEIGHT HIGH DAMPING POROUS  
9 METAL/PHTHALONITRILE COMPOSITES, filed on even date herewith, the  
10 entirety of which is incorporated herein by reference.  
11  
12

13 Obviously, many modifications and variations of the present  
14 invention are possible in light of the above teachings. It is  
15 therefore to be understood that, within the scope of the appended  
16 claims, the invention may be practiced otherwise than as  
specifically described.

**Claims**

What is claimed is:

1. An acoustically damping composite article, comprising a metal foam, said metal foam having an open cell structure, impregnated with a solid, bulk non-elastomeric polymer.
2. The composite article of claim 1, wherein said metal is selected from the group consisting of aluminum, aluminum base alloys, titanium, titanium base alloys, nickel, nickel base alloys, copper, copper base alloys, iron, iron base alloys, zinc, zinc base alloys, lead, lead base alloys, silver, silver base alloys, gold, gold base alloys, platinum, platinum base alloys, tantalum, and tantalum base alloys.
3. The composite article of claim 1, wherein said polymer is selected from the groups consisting of epoxies, acrylics, hardened silicones, polyurethanes, polyimides, polyvinyls, polycarbonates, hardened natural rubbers, hardened synthetic rubbers, phenolics, polyolefins, polyamides, polyesters, fluoropolymers, poly(phenylene ether ketones), poly(phenylene ether sulfones), poly(phenylene sulfides) and melamine-formaldehyde resins.

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4. The composite article of claim 1, wherein said metal is an aluminum base alloy foam.

5. The composite article of claim 1, wherein said metal is a copper foam or a copper base alloy foam.

6. The composite article of claim 1, wherein said metal is a zinc foam or a zinc base alloy foam.

7. The composite article of claim 3, wherein said metal is an aluminum foam or an aluminum base alloy foam.

8. The composite article of claim 3, wherein said metal is a copper foam or a copper base alloy foam.

9. The composite article of claim 3, wherein said metal is a zinc foam or a zinc base alloy foam.

10. The composite article of claim 3, wherein said metal is a titanium foam or a titanium base alloy foam.

11. The composite article of claim 1, wherein said polymer is an epoxy.

12. The composite article of claim 1, wherein said polymer is an

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acrylic.

13. The composite article of claim 1, wherein said polymer is a hardened silicone rubber.

14. The composite article of claim 1, wherein said polymer is a hardened natural rubber.

15. The composite article of claim 1, wherein said polymer is a hardened synthetic non-silicone rubber.

16. The composite article of claim 1, wherein said polymer is a phenolic.

17. The composite article of claim 1, wherein said cells have a locally uniform diameter.

18. The composite article of claim 17, wherein said metal foam has a gradation of pores sizes in at least one direction along the metal foam.

19. A composite article according to claim 1, wherein said composite article is in the form of a sheet.

20. A laminate comprising a plurality of sheets according to claim

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19 bonded together.

21. An acoustically damping composite article, comprising a metal foam, said metal foam having an open cell structure, impregnated with solid bulk polymer, said metal foam being, along its smallest dimensions, no smaller than 3 times the average diameter of said cells..

22. A method of forming a composite comprising the steps of:  
    impregnating a metal foam, said metal foam having an open cell structure, with a resin component; and  
    converting said resin component, within said cells, to a bulk solid, non-elastomeric polymerized resin.

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#### ABSTRACT

Metal foams are impregnated with resins. The metal foam/polymer composite formed upon curing has excellent acoustic dampening and structural properties. Foams of various metals, such as aluminum, titanium, nickel, copper, iron, zinc, lead, silver, gold, platinum, tantalum, and alloys based on these metals, may be used. The polymer component may be any polymeric resin, for example, epoxy, natural rubber, acrylic, or phenolic.

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SERIALIZED FILED

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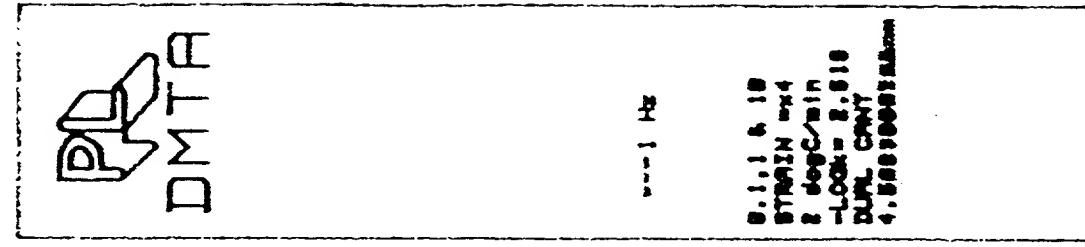
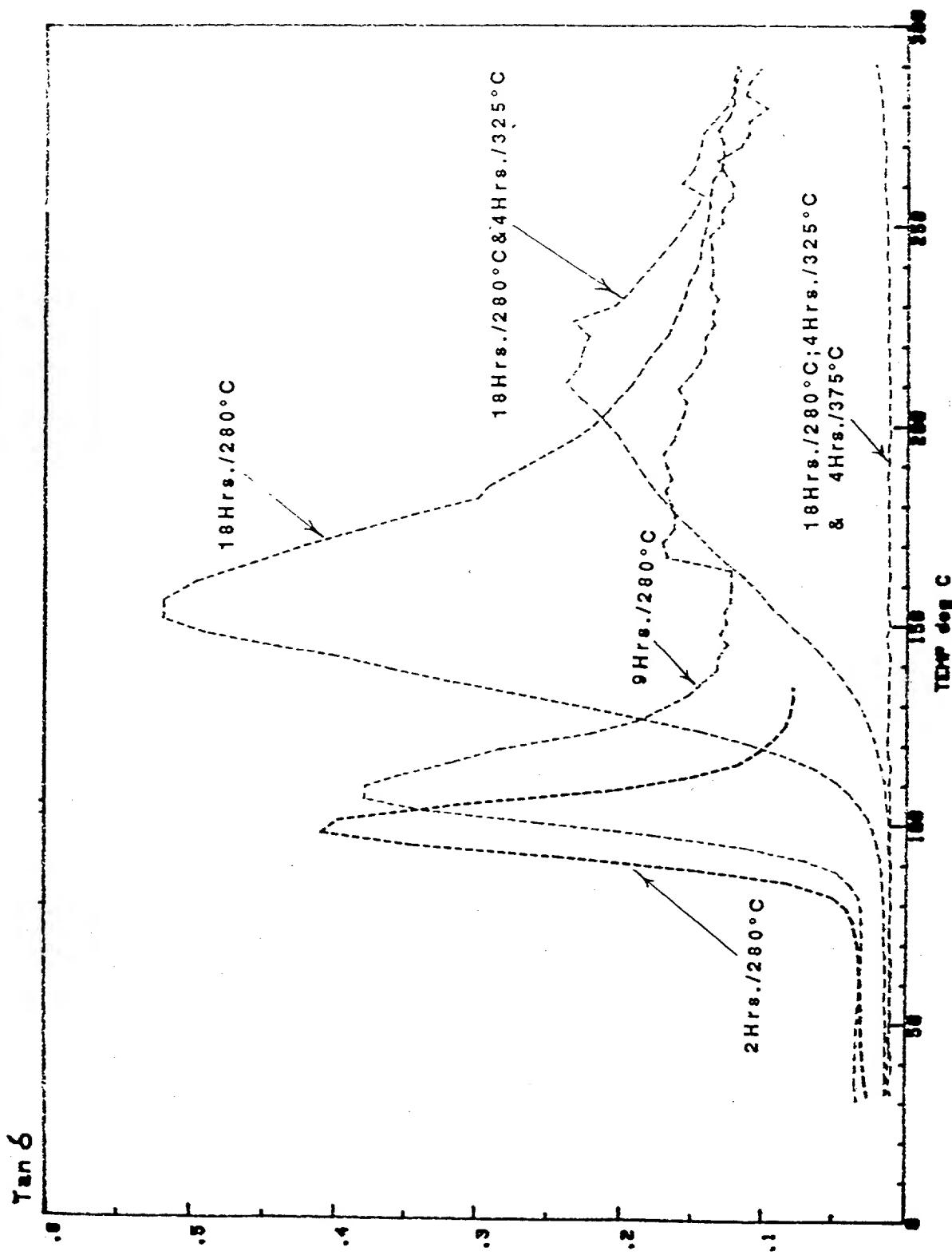
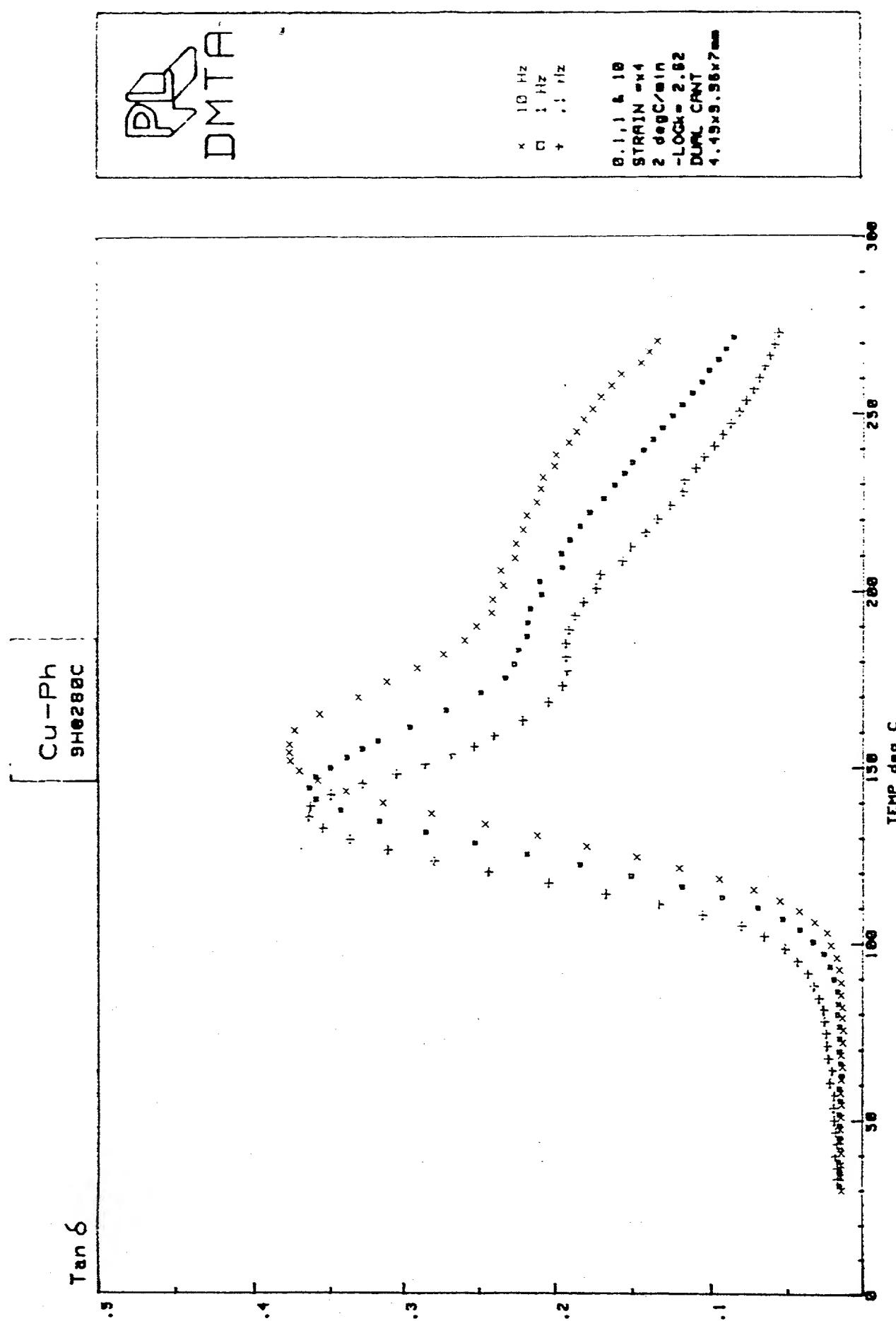


Fig 1

Fig. 2



200 180 160 140 120 100 80 60

Cu-Ph  
16H0280C; 4H0325C

Tan δ

.28

.24

.2

.16

.12

.08

.04

0

TEMP deg C

360  
320  
280  
240  
200  
160  
120  
80

PLA DMTA

0.1, 1 & 10  
STRAIN = x4  
2 degC/min  
V-LOCK = 2.682  
DUAL CRNT  
4.21x9.79x7mm

10 Hz  
1 Hz  
0.1 Hz

Fig. 3

Fig. 4

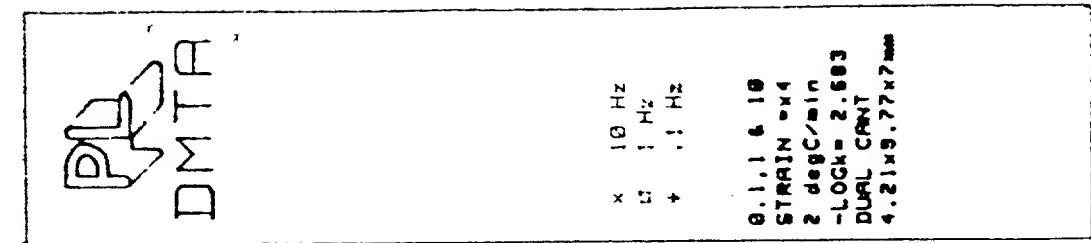
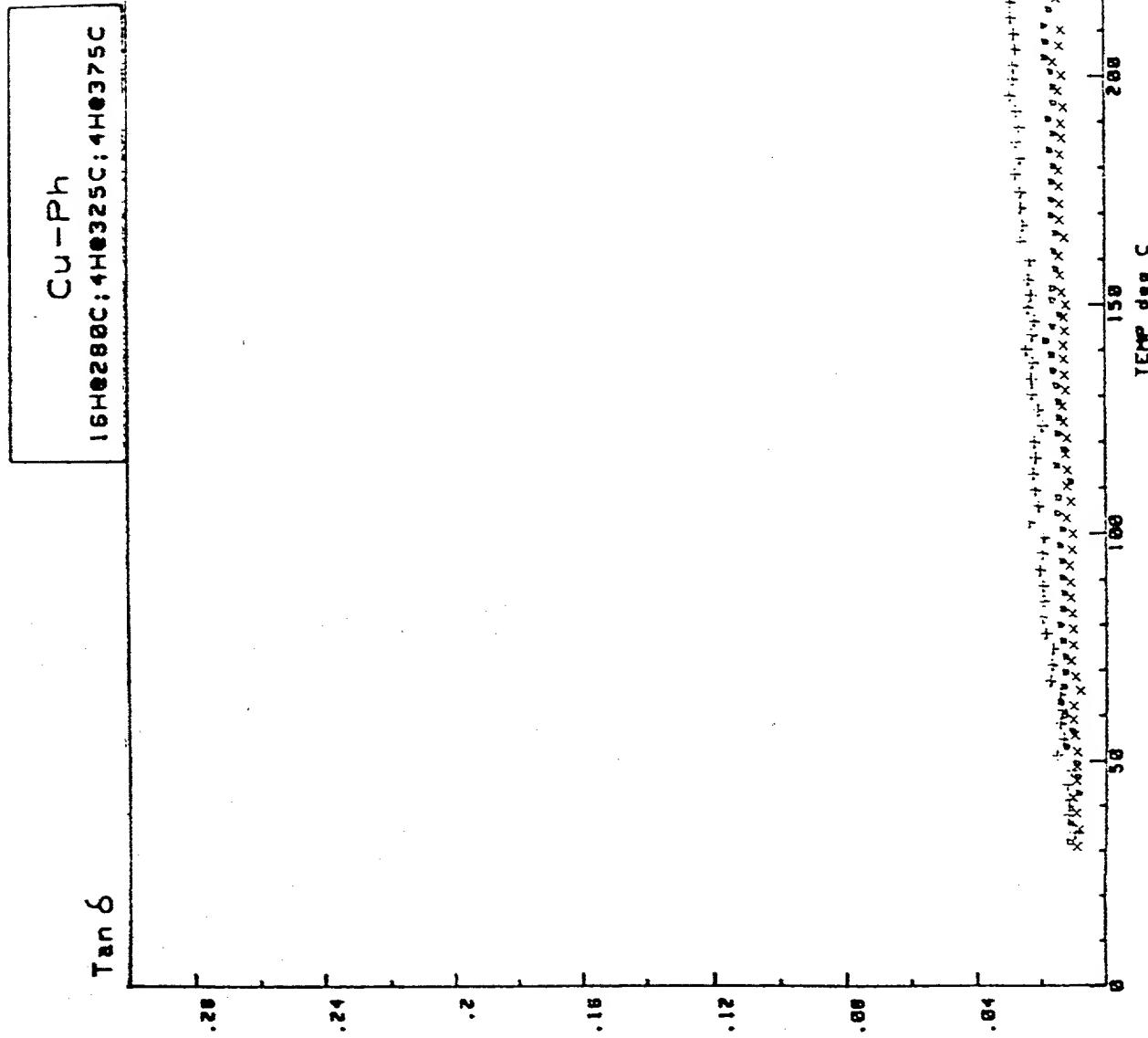
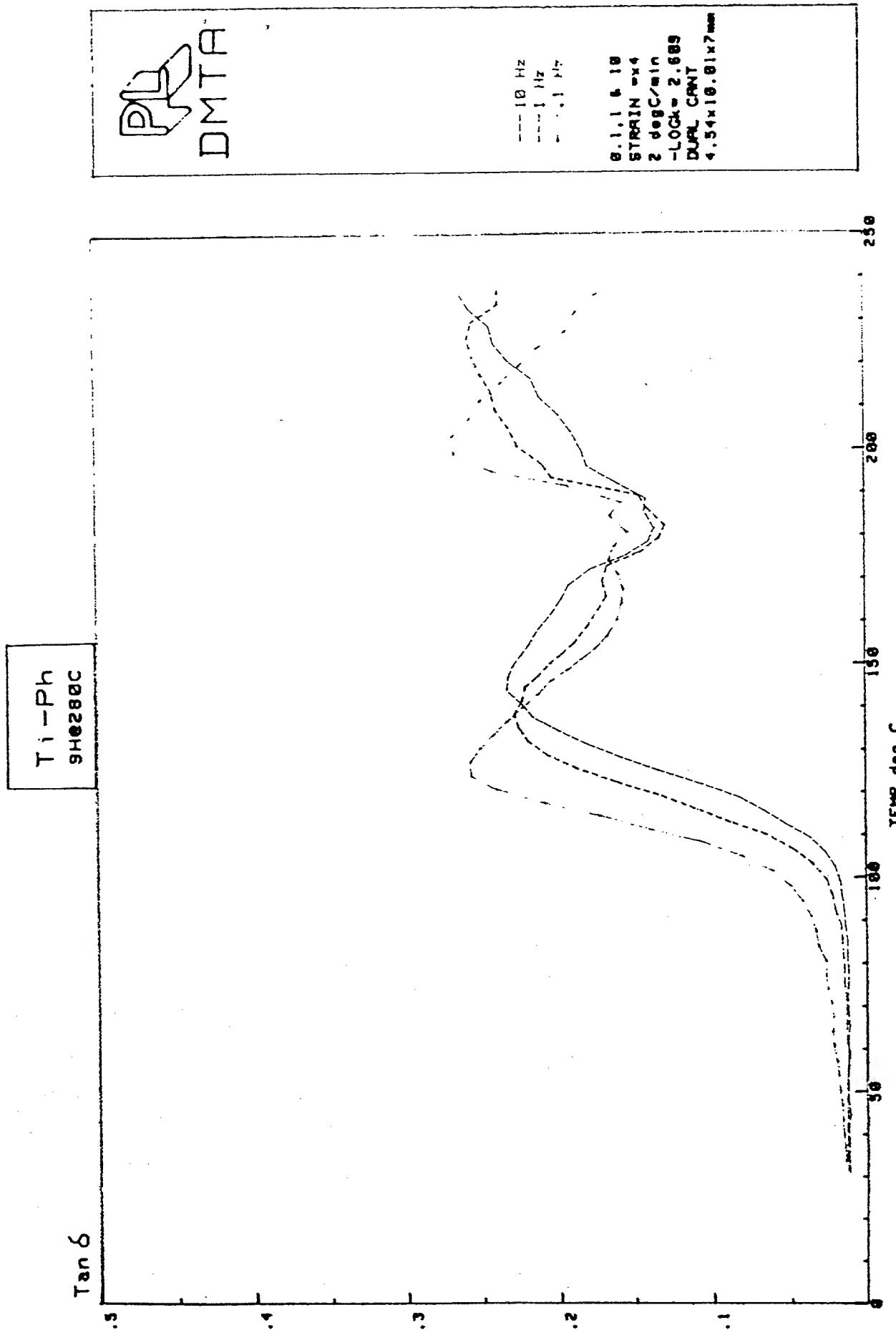


Fig. 5



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Ph/T<sub>1</sub> Composite  
18H0200C14H0325C

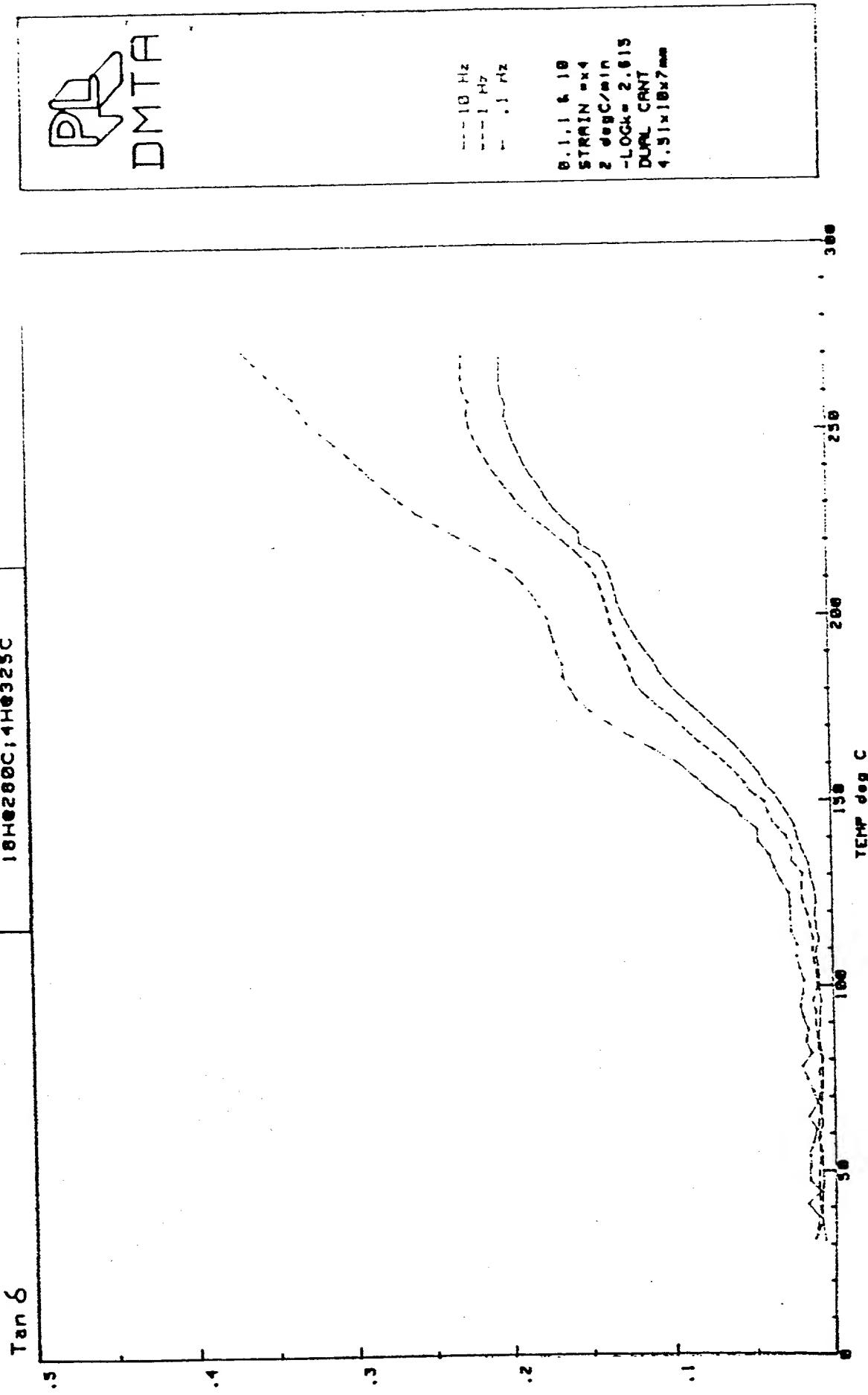
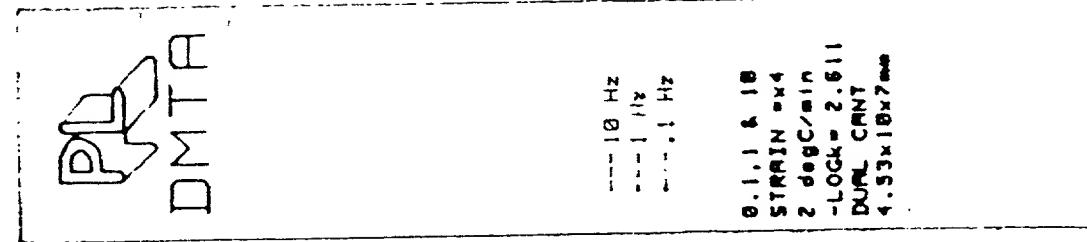
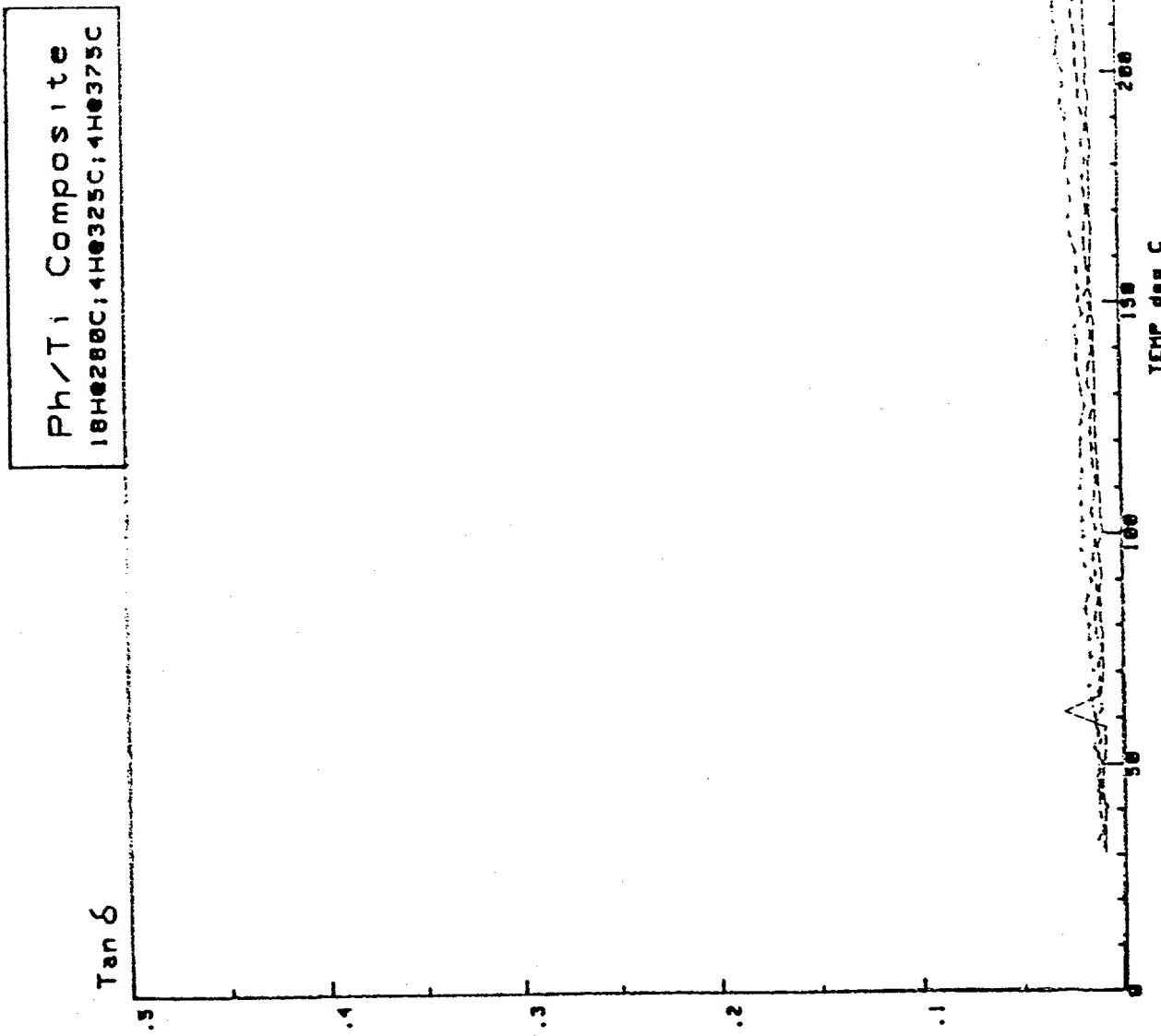


Fig. 6

Fig.7



0.1.1 & 1B  
STRAIN = x4  
2 degC/min  
-LOCK = 2.611  
DUML CRNT  
4.53x10x7mm

1 Hz  
2 Hz  
10 Hz

26 SEPTEMBER 1986

Ph/Zn Composite  
18H280C; 4H0325C

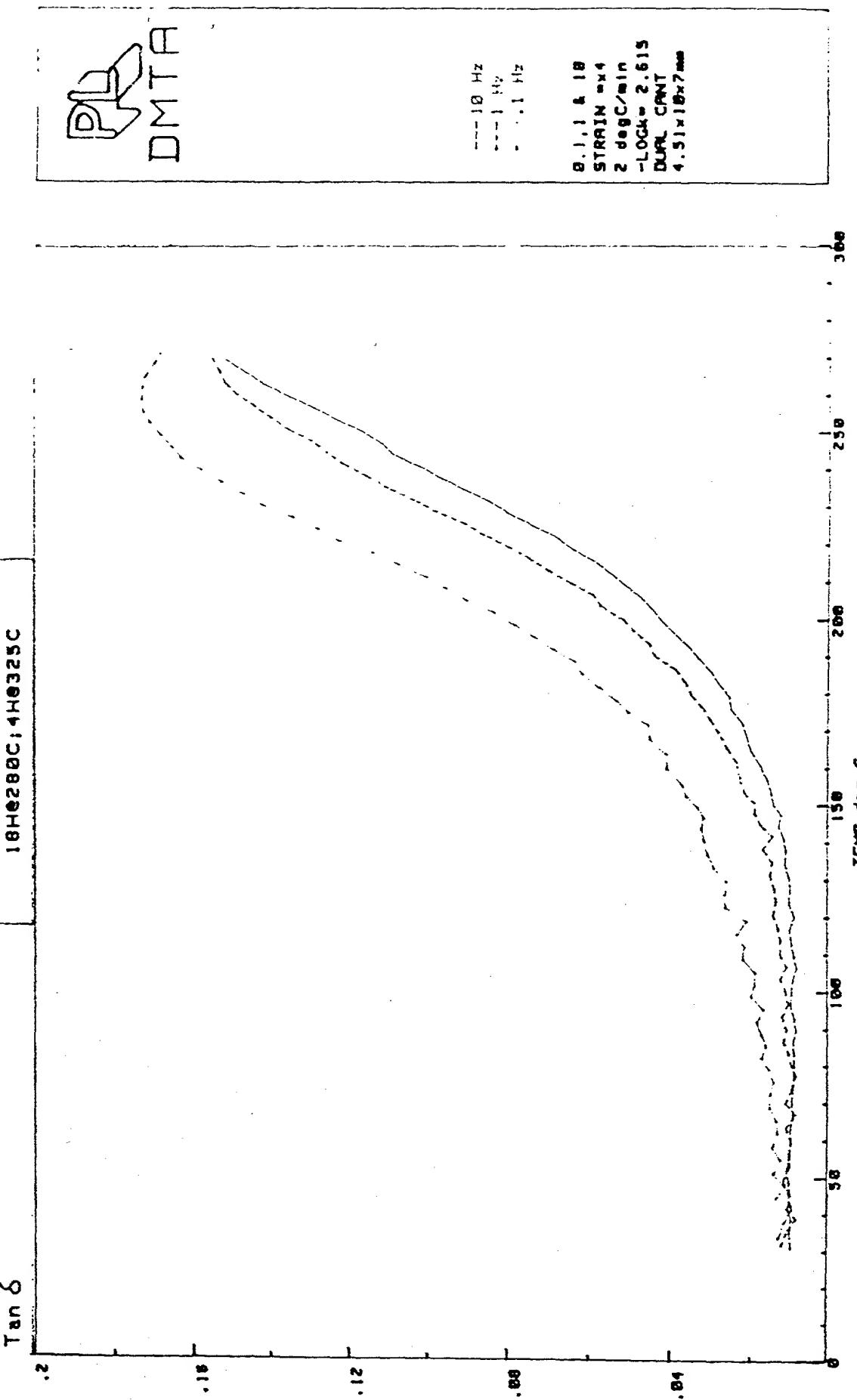


Fig. 8

Ph/Zn Composite  
18H0280C; 1H0325C; 1H0375C

Tan δ

2

1

0

-1

-2

11

12

13

14

15

16

17

18

19

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21

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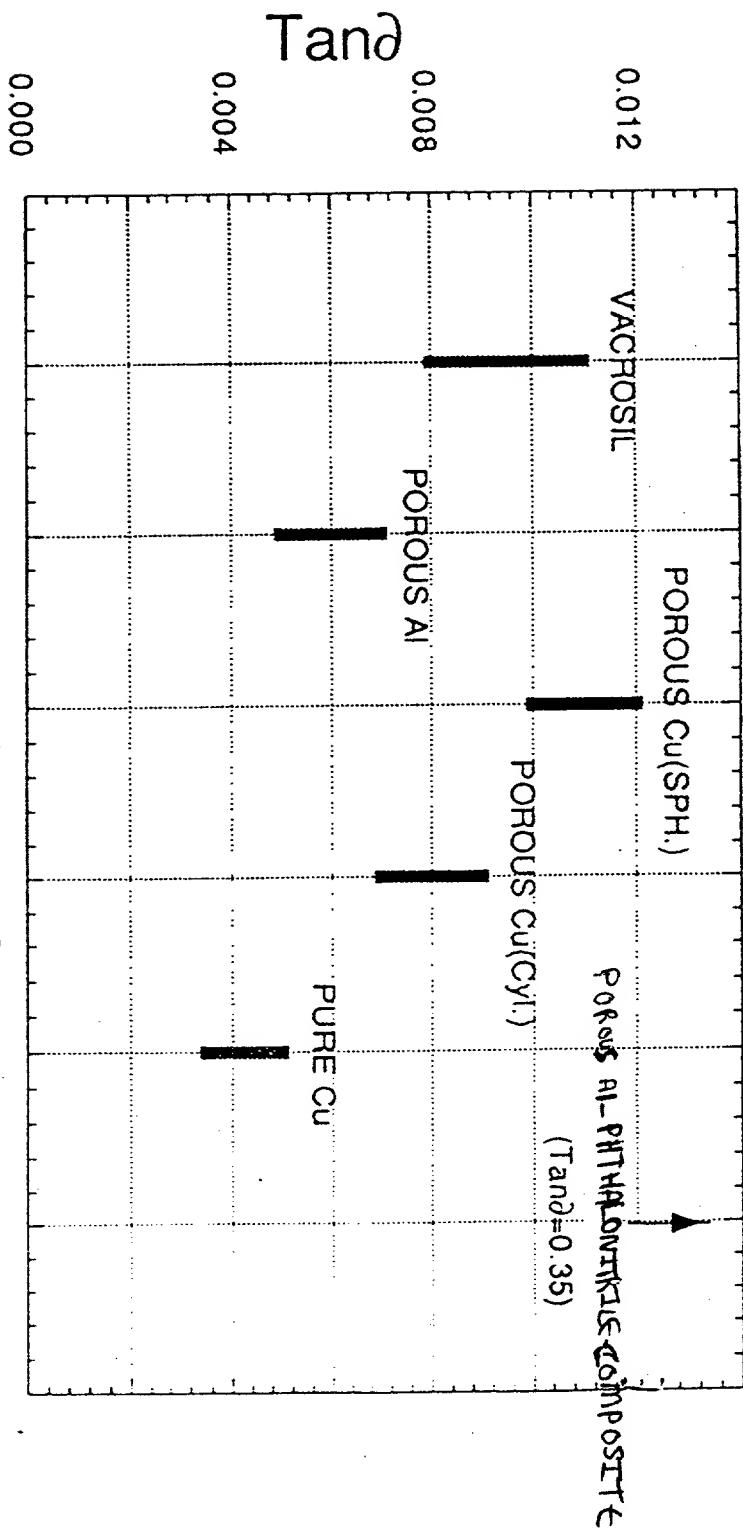
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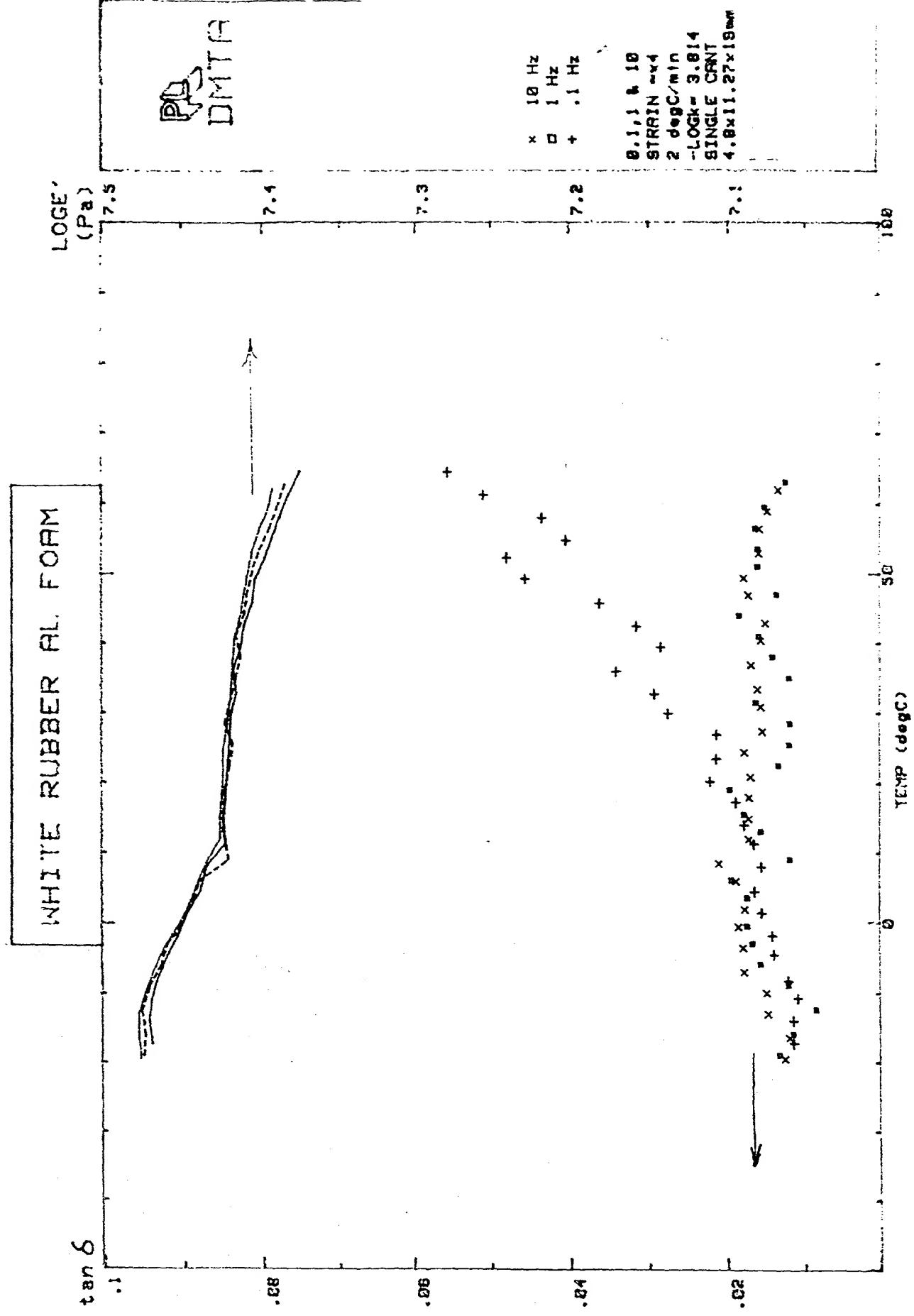
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## Tan $\delta$ Over Frequency Range of 0.1 to 10 Hz



## MATERIALS

Fig. II



20 18 16 14 12 10 8 6 4 2 0

RED RUBBER AL FORM

Tan δ

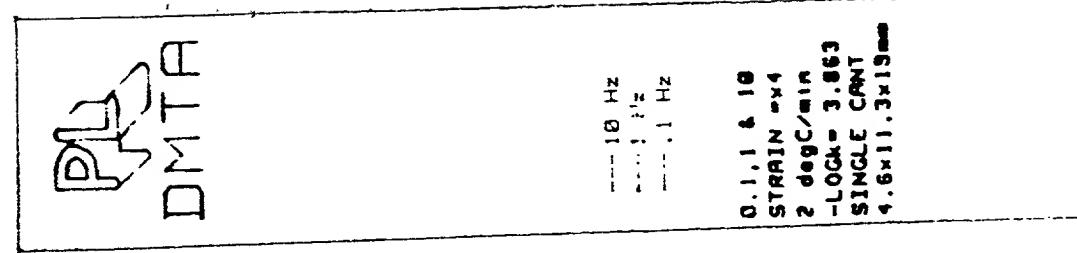
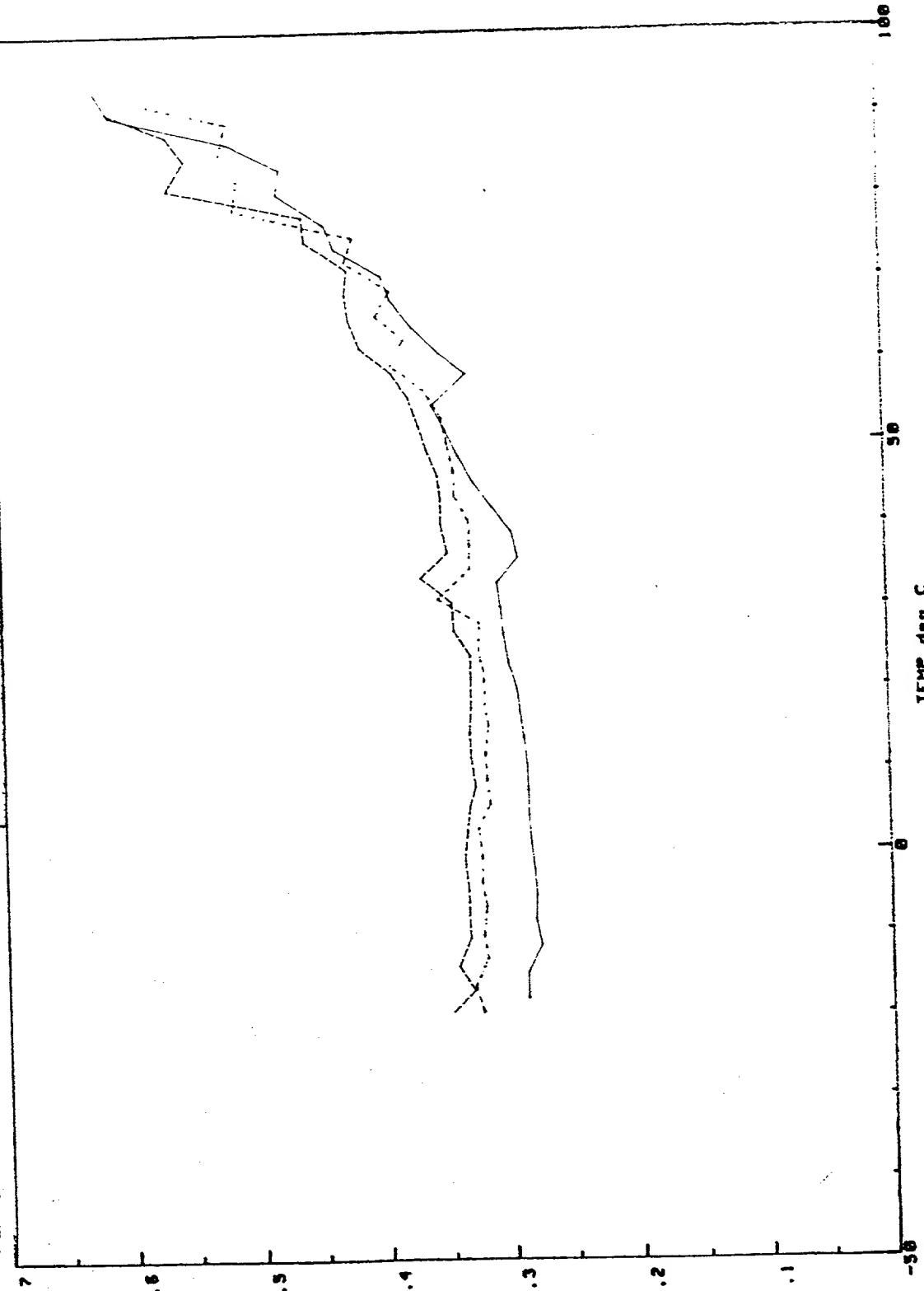


Fig. 12

Fig. 13

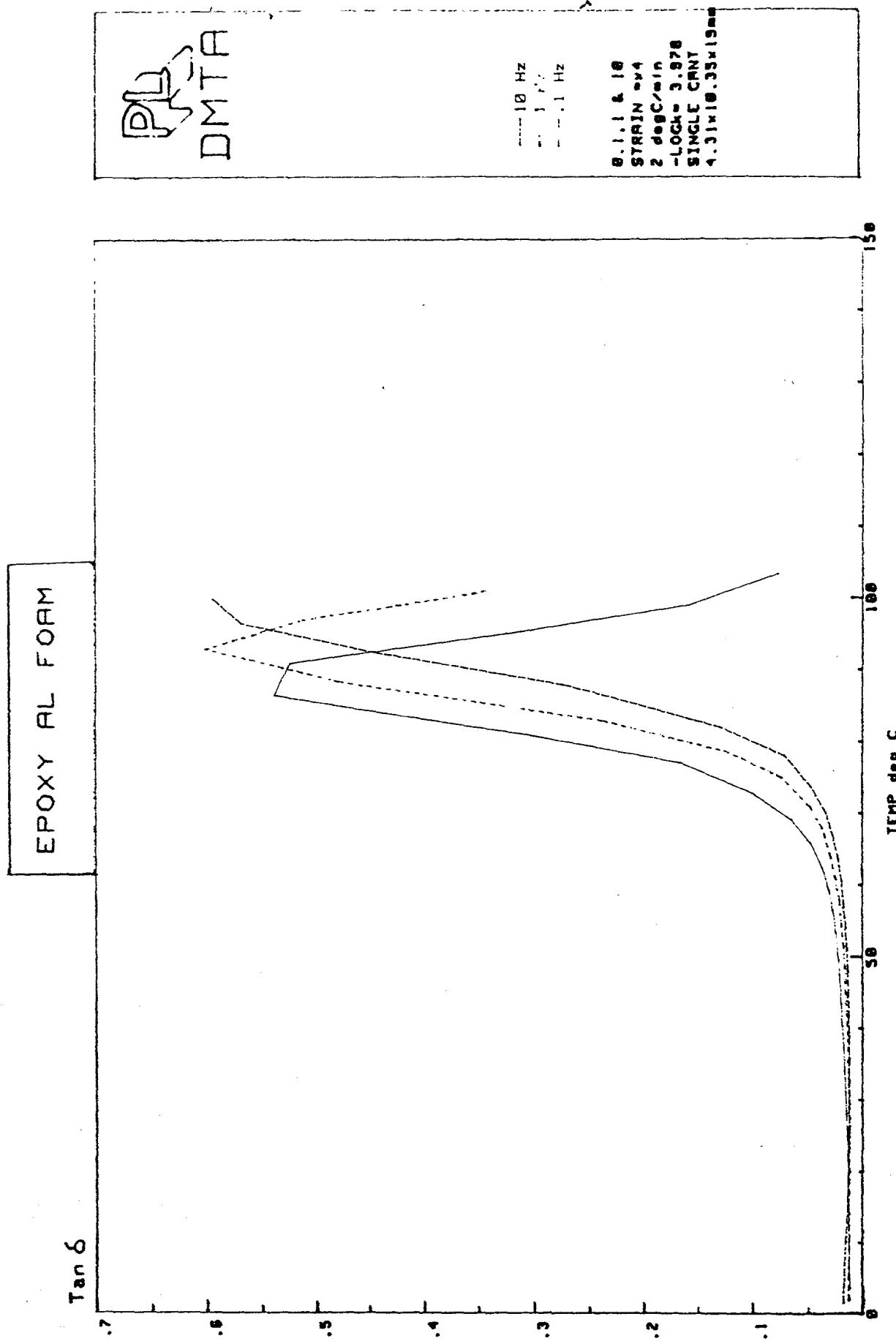
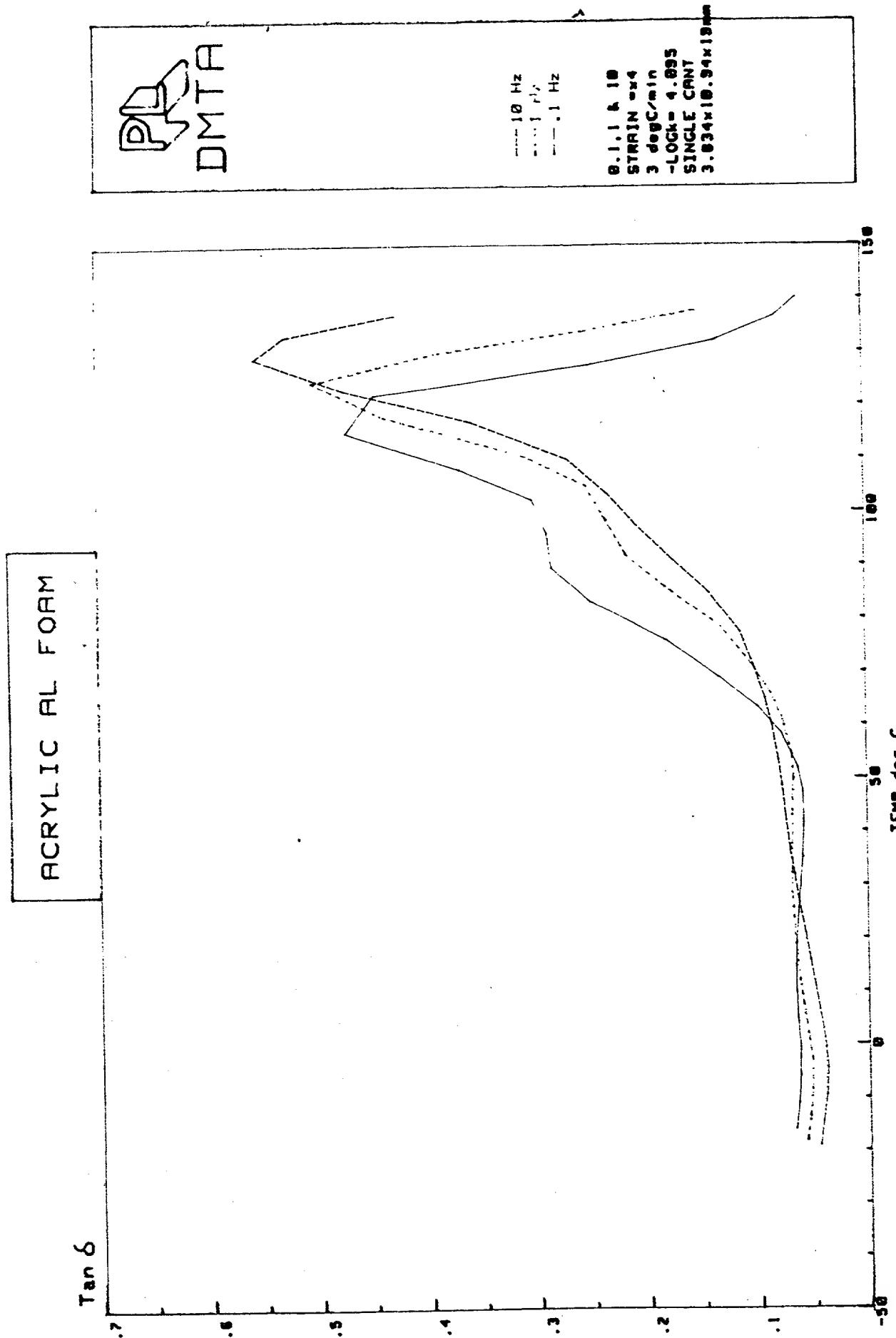


Fig. 14



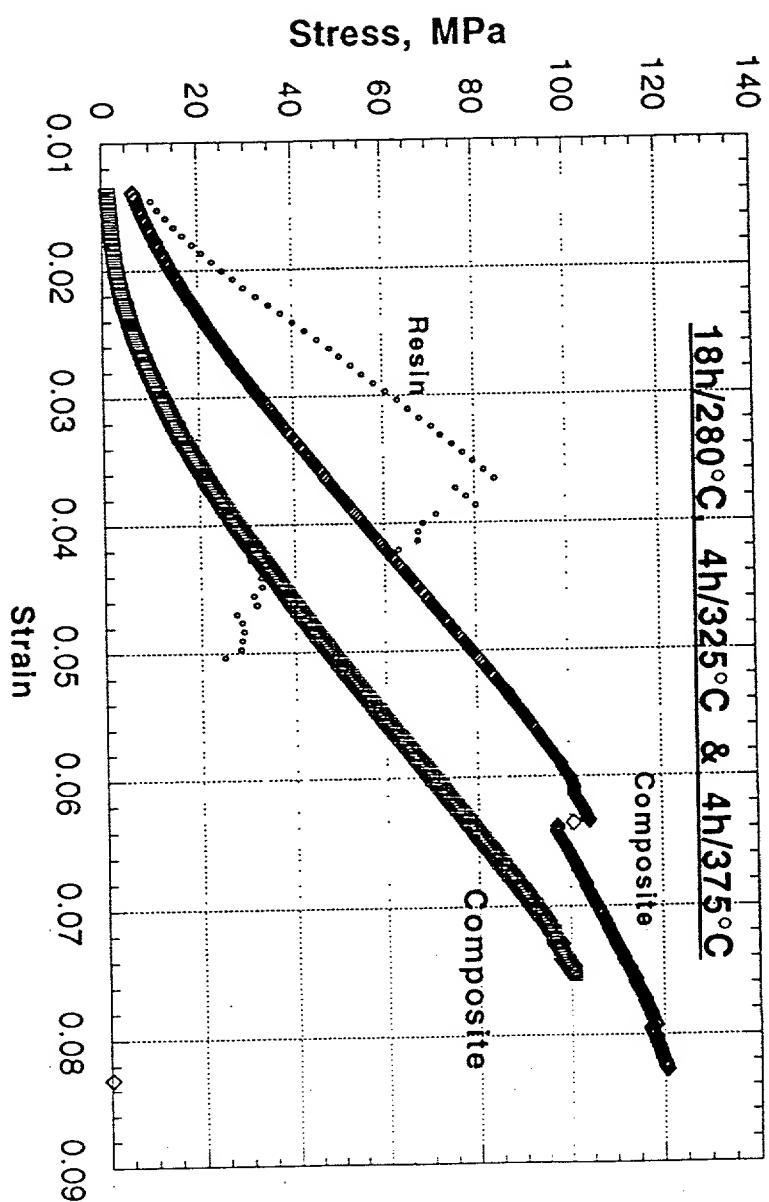
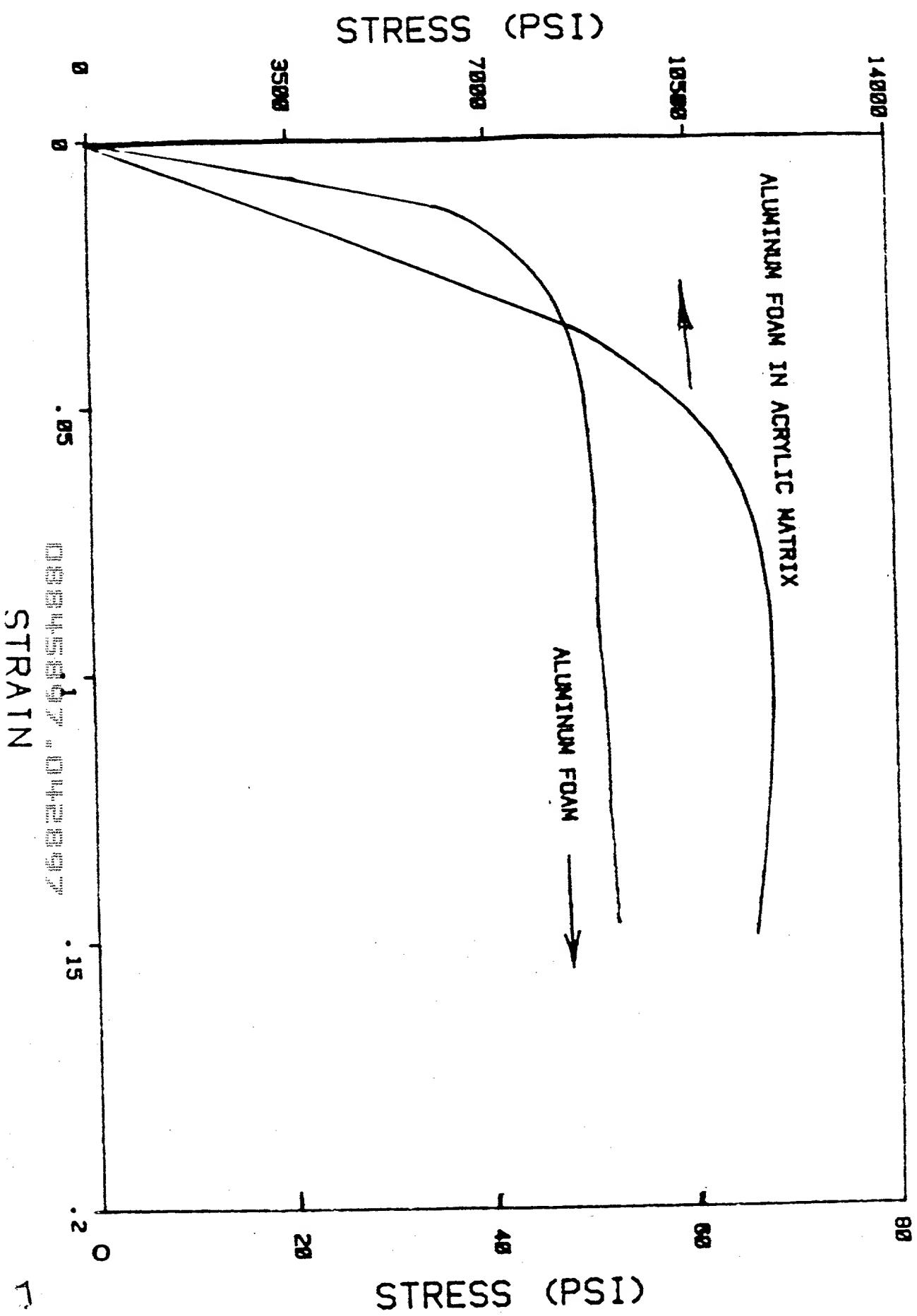


Fig. 15

STRESS VS STRAIN



Declaration and Power of Attorney For Patent Application

As a below named inventor, I hereby declare that:

My residence, post office address and citizenship are as stated below next to my name.

I believe that I am the original, first and sole inventor (if only one name is listed below) or an original, first and joint inventor (if plural names are listed below) of the subject matter which is claimed and for which a patent is sought on the invention entitled: "POROUS METAL/ORGANIC POLYMERIC COMPOSITES" the specification of which:

(check one)  is attached hereto. [ ] was filed on \_\_\_\_\_ as Application Serial No. \_\_\_\_\_ and was amended on \_\_\_\_\_.

I hereby state that I have reviewed and understand the contents of the above identified specification, including the claims, as amended by any amendment referred to above.

I acknowledge the duty to disclose information which is material to the examination of this application in accordance with Title 37, Code of Federal Regulations, §1.56(a).

I hereby claim foreign priority benefits under Title 35, United States Code, §119 of any foreign applications for patent or inventor's certificate listed below and have also identified below any foreign application for patent or inventor's certificate having a filing date before that of the application on which priority is claimed:

Prior Foreign Applications (Number)	(Country)	(Day/Month/Year Filed)	Priority Claimed Yes	No
--	-----------	------------------------	-------------------------	----

I hereby claim the benefit under Title 35, United States Code, §120 of any United States applications listed below and, insofar as the subject matter of each of the claims of this application is not disclosed in the prior United States application in the manner provided by the first paragraph of title 35, United States Code, §112, I acknowledge the duty to disclose material information as defined in Title 37, Code of Federal Regulations, §1.56(a) which occurred between the filing date of the prior application and the national or PCT international filing date of this application:

(Application Serial No.)	(Filing Date)	(Status) (patented, pending, abandoned)
--------------------------	---------------	--

**POWER OF ATTORNEY:** As a named inventor, I hereby appoint the following attorneys/and/or agent/s/ to prosecute this application and transact all business in the Patent and Trademark Office connected therewith, and hereby certify that the Government of the United States has the irrevocable right to prosecute this application:

Thomas E. McDonnell,  
Reg., No. 26950

Barry A. Edelberg  
Reg. No. 31012

SEND CORRESPONDENCE TO:  
Associate Counsel (Patents), Code 3008.2  
Naval Research Laboratory  
Washington, D.C. 20375-5000

DIRECT TELEPHONE CALLS TO:  
Barry A. Edelberg  
(202)-404-1558

I hereby declare that all statements made herein of own knowledge are true and that all statements made on information and belief are believed to be true; and further that these statements were made with the knowledge that willful false statements and the like so made are punishable by fine or imprisonment, or both, under Section 1001 of Title 18 of the United States Code, and that such willful false statements may jeopardize the validity of the application or any patent issued thereon.

Full name of first inventor: M. Ashraf Imam

Inventor's signature: M. Ashraf Imam Date 4/28/97

Residence: Great Falls, Va.

Citizenship: USA

Post Office Address: 1159 Mill Garden Ct., Great Falls, Va., 22066

Full name of second inventor: Shakti B. Path

Inventor's signature: Shakti B. Path Date 4/28/97

Residence: Oakton, Va.

Citizenship: USA

Post Office Address: 10908 Timbermill Ct., Oakton, Va., 22124

Declaration and Power of Attorney For Patent Application

As a below named inventor, I hereby declare that:

My residence, post office address and citizenship are as stated below next to my name.

I believe that I am the original, first and sole inventor (if only one name is listed below) or an original, first and joint inventor (if plural names are listed below) of the subject matter which is claimed and for which a patent is sought on the invention entitled: "POROUS METAL/ORGANIC POLYMERIC COMPOSITES" the specification of which:

(check one)  is attached hereto. [] was filed on \_\_\_\_\_ as Application Serial No. \_\_\_\_\_ and was amended on \_\_\_\_\_.

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Prior Foreign Applications	Priority Claimed			
<u>(Number)</u>	<u>(Country)</u>	<u>(Day/Month/Year Filed)</u>	<u>Yes</u>	<u>No</u>

I hereby claim the benefit under Title 35, United States Code, §120 of any United States applications listed below and, insofar as the subject matter of each of the claims of this application is not disclosed in the prior United States application in the manner provided by the first paragraph of title 35, United States Code, §112, I acknowledge the duty to disclose material information as defined in Title 37, Code of Federal Regulations, §1.56(a) which occurred between the filing date of the prior application and the national or PCT international filing date of this application:

(Application Serial No.)	(Filing Date)	(Status)
		(patented, pending, abandoned)

POWER OF ATTORNEY: As a named inventor, I hereby appoint the following attorneys/and/or agent/s/ to prosecute this application and transact all business in the Patent and Trademark Office connected therewith, and hereby certify that the Government of the United States has the irrevocable right to prosecute this application:

Thomas E. McDonnell,  
Reg., No. 26950

Barry A. Edelberg  
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Barry A. Edelberg  
(202)-404-1558

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Full name of third inventor: Teddy M. Keller

Inventor's signature: Teddy M. Keller

Date 4/28/97

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